

Quantum Mechanical Simulation of Nano-composite Magnets on CRAY XT3

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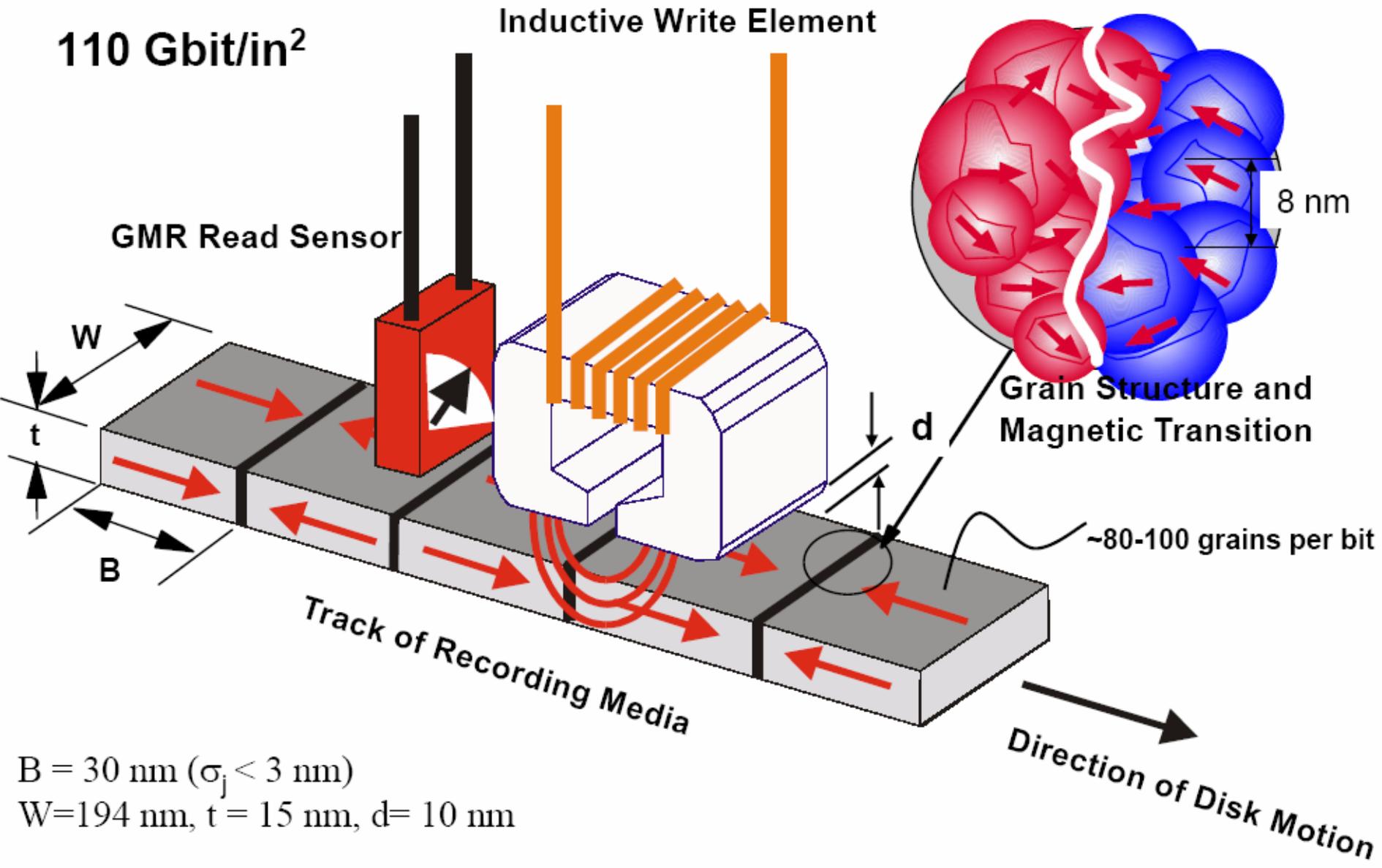
Magnetic Nanoparticle Composites

Nanoscale magnetic crystallites formed in an amorphous or crystalline matrix or on a surface

- ❑ Properties:
 - Atomic scale effects play dominant role.
 - Physical parameters can be adjusted selectively.
- ❑ The advantages: an increased sensitivity to detect changes of magnetic field and a higher working temperature range.
- ❑ Potential applications: sensors, data storage, MRI contrast enhancing reagents, smart drug delivery, etc.

Science of Disk Drives

110 Gbit/in²



$B = 30\text{ nm}$ ($\sigma_j < 3\text{ nm}$)

$W = 194\text{ nm}$, $t = 15\text{ nm}$, $d = 10\text{ nm}$

Pushing the Limit

□ Present disk drives ($\sim 100 \text{ Gb/in}^2$)

Use a metallic thin-film medium whose magnetic grains (~ 100 grains/bit), acting like an array of permanent magnet particles, are partially isolated from one another by a nonmagnetic chromium-rich alloy.

□ Superparamagnetic Limit ($\sim 100\text{-}200 \text{ Gb/in}^2$)

A factor of 2 decrease in grain diameter \rightarrow a factor of 8 decrease of grain volume \rightarrow a factor of 8 decrease of magnetic energy stored in each grain \rightarrow a change of the magnetic moment reversal time, due to thermal fluctuation, from 100 years to as much as 100 nanoseconds.

□ Future magnetic data storage ($\sim 1 \text{ Tb/in}^2$)

- High magnetic anisotropy multilayers (Co/Pd, or FePt film in $L1_0$ phase)
- Perpendicular recording
- Magnetic nanoparticle composites with high magnetic anisotropy and 1 particle/bit:
FePt, CoPt, etc.

Theoretical Approach

□ Empirical

- Model potentials for inter-atomic interactions
- Less accuracy and poor predictability

□ Semi-empirical

- Parameters in the model potential are determined quantum mechanically
- Better accuracy and reasonable predictability

□ Quantum mechanical (*ab initio*)

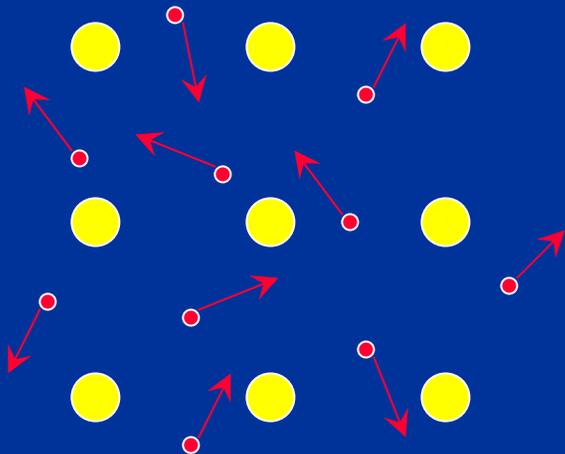
- No models
- The best accuracy and predictability

Quantum Mechanical Solution of Materials Science Problems

Electron: ●

Nucleus: ●

Many-electron problem

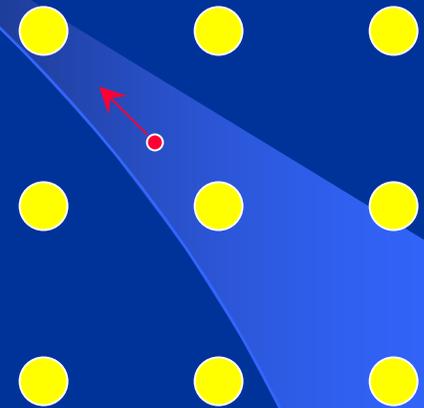


electron-electron interaction
 electron-nucleus interaction
 many-electron Schrödinger equation

Density Functional Theory



One-electron problem



non-interacting electrons move in an
 effective potential: $V_{\text{eff}}[\rho]$
 one-electron Schrödinger equation

$$\left(-\frac{\hbar^2}{2m_e} \nabla^2 + e^2 \int_{\infty} \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3\vec{r}' - e^2 \sum_{\vec{R}_n} \frac{Z_n}{|\vec{r} - \vec{R}_n|} + V_{\text{xc}}[\rho] \right) \Psi_{\alpha}(\vec{r}) = \varepsilon_{\alpha} \Psi_{\alpha}(\vec{r})$$

Self-consistent Process

$V_{\text{eff}}(\vec{r})$, lattice parameters.

One-electron Schrödinger Equation

$$[-\nabla^2 + V_{\text{eff}}(\vec{r})] \cdot \Psi_{\alpha}(\vec{r}) = \varepsilon_{\alpha} \Psi_{\alpha}(\vec{r})$$

$E[\rho]$, etc.

$V_{\text{eff}}(\vec{r})$

$$\rho(\vec{r}) = \sum_{\alpha} |\Psi_{\alpha}(\vec{r})|^2$$

LDA Potential

$$V_{\text{eff}}(\vec{r}) = e^2 \int_{\infty} \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3\vec{r}' - e^2 \sum_{\vec{R}_n} \frac{Z_n}{|\vec{r} - \vec{R}_n|} + V_{\text{XC}}^{\text{LDA}}[\rho]$$

$\rho(\vec{r})$, lattice parameters.

Local Density Approximation (LDA)
makes calculation of $V_{\text{XC}}[\rho]$ feasible.

Computational Challenges

- ❑ A spherical particle of 3 nm in diameter ~ 2000 atoms
- ❑ Bottlenecks of conventional *ab initio* electronic structure methods
 - N^3 scaling in computational requirement
 - N^2 scaling in memory requirement
 - Dominated by global operations
 - Lack of efficient parallel implementation scheme
 - Simulation size is limited to 100s atoms

Locally Self-consistent Multiple Scattering (LSMS) Method

- Real space multiple scattering approach

$$\text{Electron Density} \quad \longrightarrow \quad \rho_i(\vec{r}) = -\frac{1}{\pi} \text{Im} \text{Tr} \int_{-\infty}^{\varepsilon_F} dz \cdot \underline{G}_i(\vec{r}, \vec{r}; z)$$

$$\text{Moment Density} \quad \longrightarrow \quad \vec{m}_i(\vec{r}) = -\frac{1}{\pi} \text{Im} \text{Tr} \int_{-\infty}^{\varepsilon_F} dz \cdot [\underline{G}_i(\vec{r}, \vec{r}; z) \cdot \underline{\vec{\sigma}}]$$

$$\text{Moment Orientation} \quad \longrightarrow \quad \vec{e}_i = \int_{\Omega_i} d^3\vec{r} \cdot \vec{m}_i(\vec{r}) / \left| \int_{\Omega_i} d^3\vec{r} \cdot \vec{m}_i(\vec{r}) \right|$$

- Solve the multiple scattering equations associated with atom i and compute the Green's function \underline{G}_i , which depends on the single scattering t -matrix and the location of the atoms in the local region.
- Order- N scaling in time and space complexity

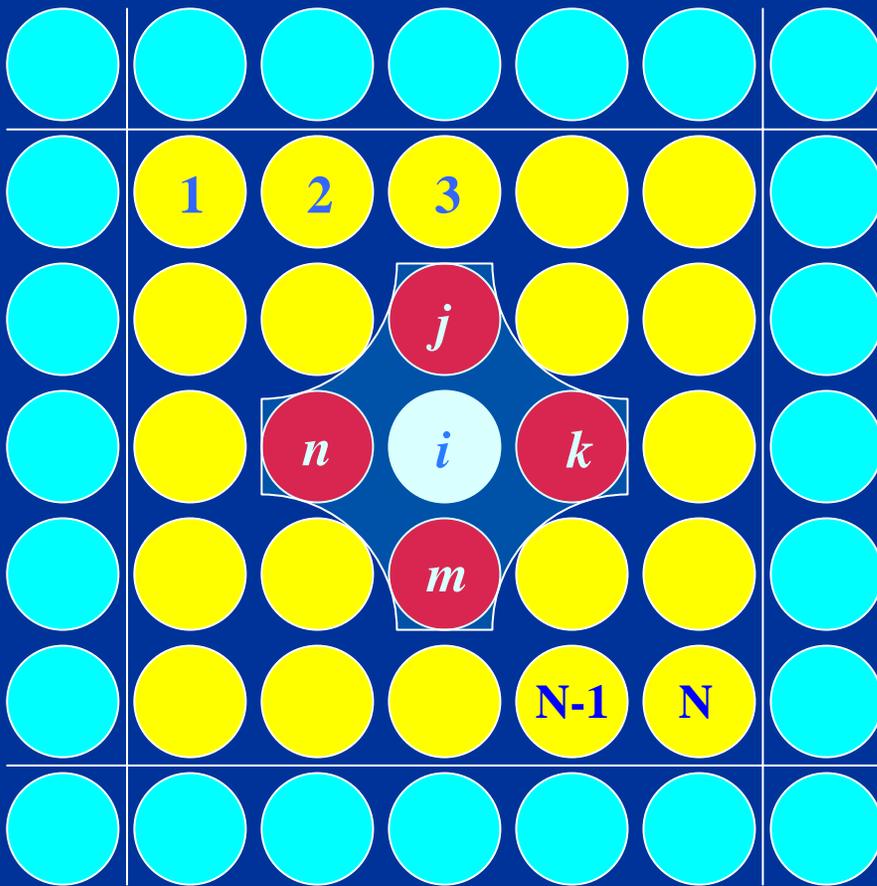
Major Floating-Point Operations

To obtain the Green's function \underline{G}_i for atom i , one needs to compute:

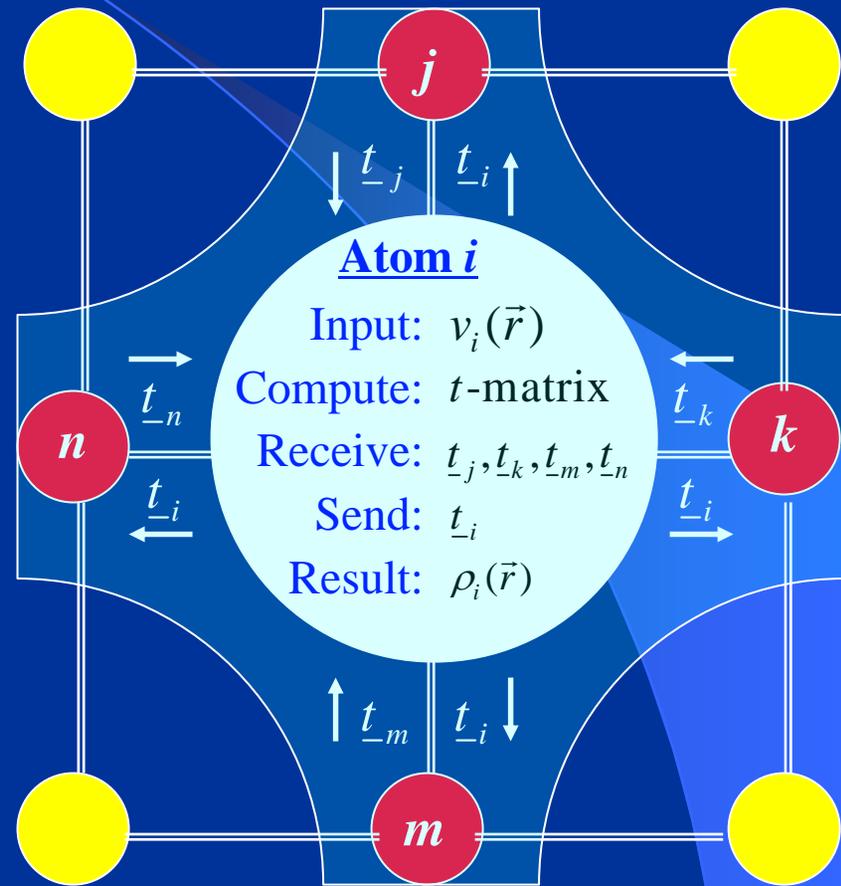
$$\underline{\tau}^{ii}(\mathcal{E}) = \left[\begin{array}{ccccc} \underline{t}_i^{-1}(\mathcal{E}) & \underline{g}_{ij}(\mathcal{E}) & \underline{g}_{ik}(\mathcal{E}) & \underline{g}_{in}(\mathcal{E}) & \underline{g}_{im}(\mathcal{E}) \\ \underline{g}_{ji}(\mathcal{E}) & \underline{t}_j^{-1}(\mathcal{E}) & \underline{g}_{jk}(\mathcal{E}) & \underline{g}_{jn}(\mathcal{E}) & \underline{g}_{jm}(\mathcal{E}) \\ \underline{g}_{ki}(\mathcal{E}) & \underline{g}_{kj}(\mathcal{E}) & \underline{t}_k^{-1}(\mathcal{E}) & \underline{g}_{kn}(\mathcal{E}) & \underline{g}_{km}(\mathcal{E}) \\ \underline{g}_{ni}(\mathcal{E}) & \underline{g}_{nj}(\mathcal{E}) & \underline{g}_{nk}(\mathcal{E}) & \underline{t}_n^{-1}(\mathcal{E}) & \underline{g}_{nm}(\mathcal{E}) \\ \underline{g}_{mi}(\mathcal{E}) & \underline{g}_{mj}(\mathcal{E}) & \underline{g}_{mk}(\mathcal{E}) & \underline{g}_{mn}(\mathcal{E}) & \underline{t}_m^{-1}(\mathcal{E}) \end{array} \right]^{-1}$$

LIZ Size $\times (l_{\max} + 1)^2$

Algorithm and Communication Pattern



N -atom Unit Cell

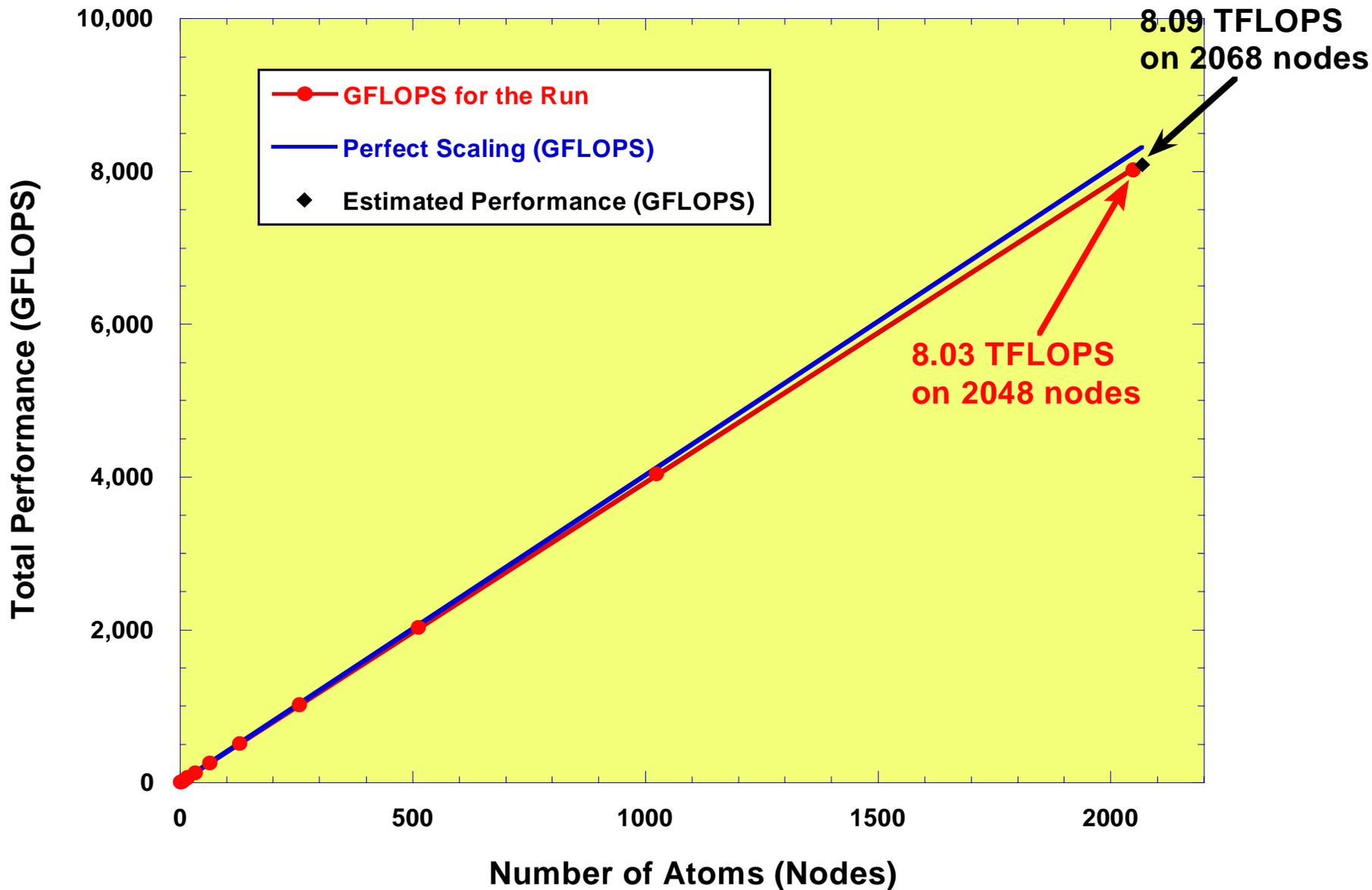


Local Interaction Zone (LIZ)

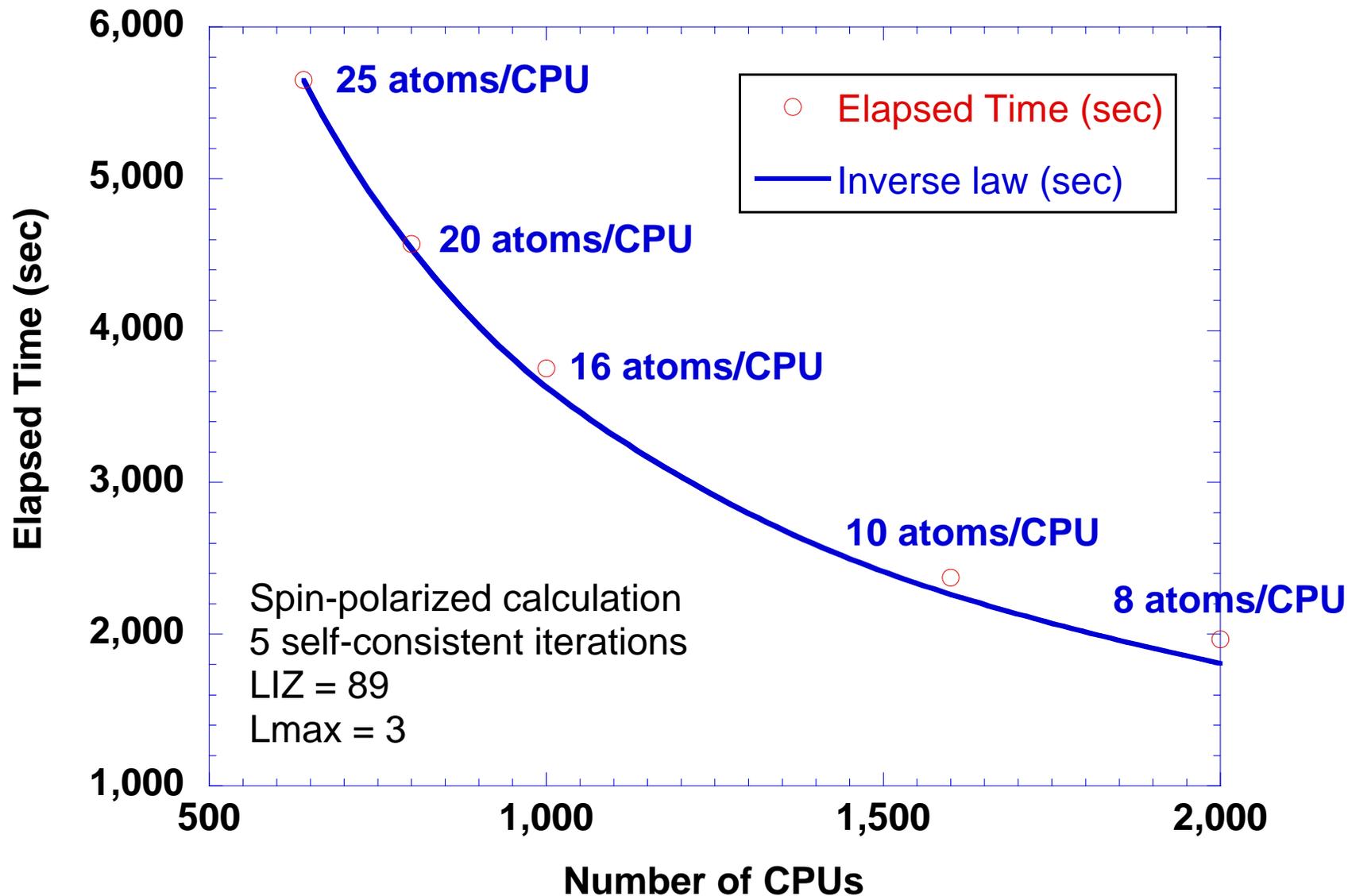
Parallel Implementation

- ❑ Intrinsic parallelism:
 - the atoms in the unit cell
 - the energy points along the complex energy contour
- ❑ Atoms are distributed evenly among CPUs, and more importantly, unlike the previous version, the code allows multiple atoms mapped onto each CPU
- ❑ One-sided communications are used for getting the t -matrix from those neighboring atoms that are mapped onto other CPUs
- ❑ Parallel I/O of data in XDR, HDF, or machine dependent binary format

LSMS Performance on Cray-XT3 (bigben) at PSC



Fe nanoparticle embedded in FeAl matrix (16,000 Fe and Al atoms per unit cell)

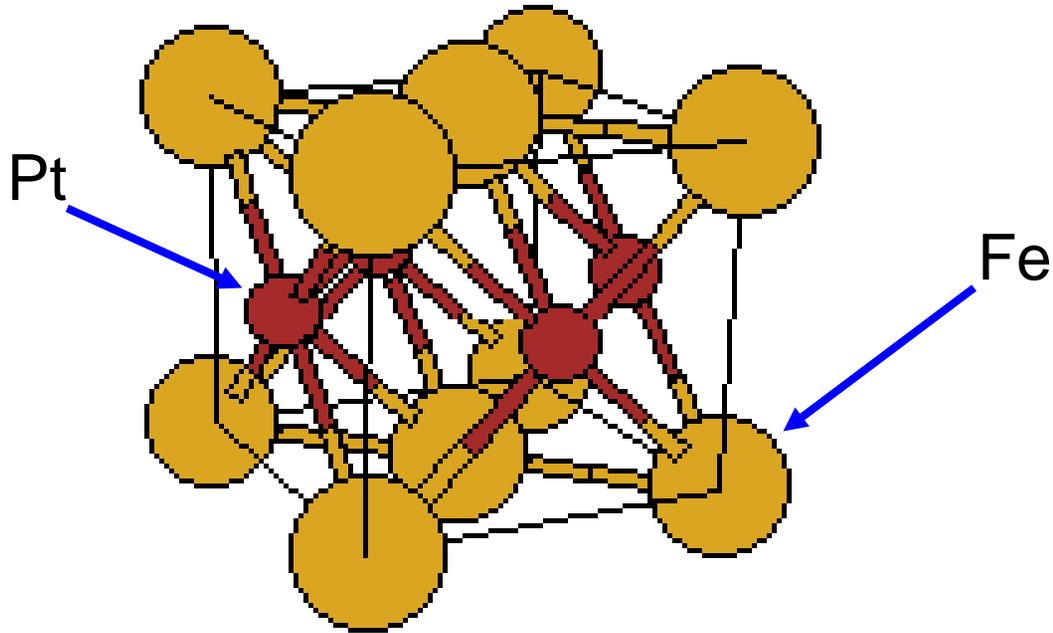


Applications of LSMS Method

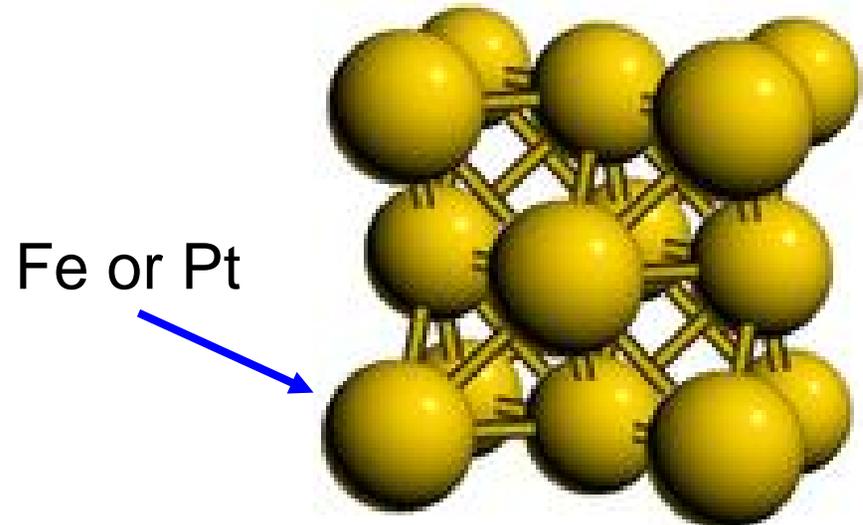
Electronic Structure of Magnetic Nanoparticles

- ❑ FePt nanoparticle ($L1_0$ structure) embedded in FePt (face-centered tetragonal structure) random alloy matrix
- ❑ No lattice mismatch ($a_0 = 3.8525\text{\AA}$ and $c_0 = 3.7133\text{\AA}$)
- ❑ Spin-polarized LSMS calculations are applied to the unit cell sample simulating the system
- ❑ The FePt nanoparticles of different sizes: 2.5nm, 3.86nm, and 5.0nm, each of which contains 711, 2,195, and 4,777 Fe and Pt atoms, respectively
- ❑ All together, there are 14,400 Fe and Pt atoms in the unit cell sample

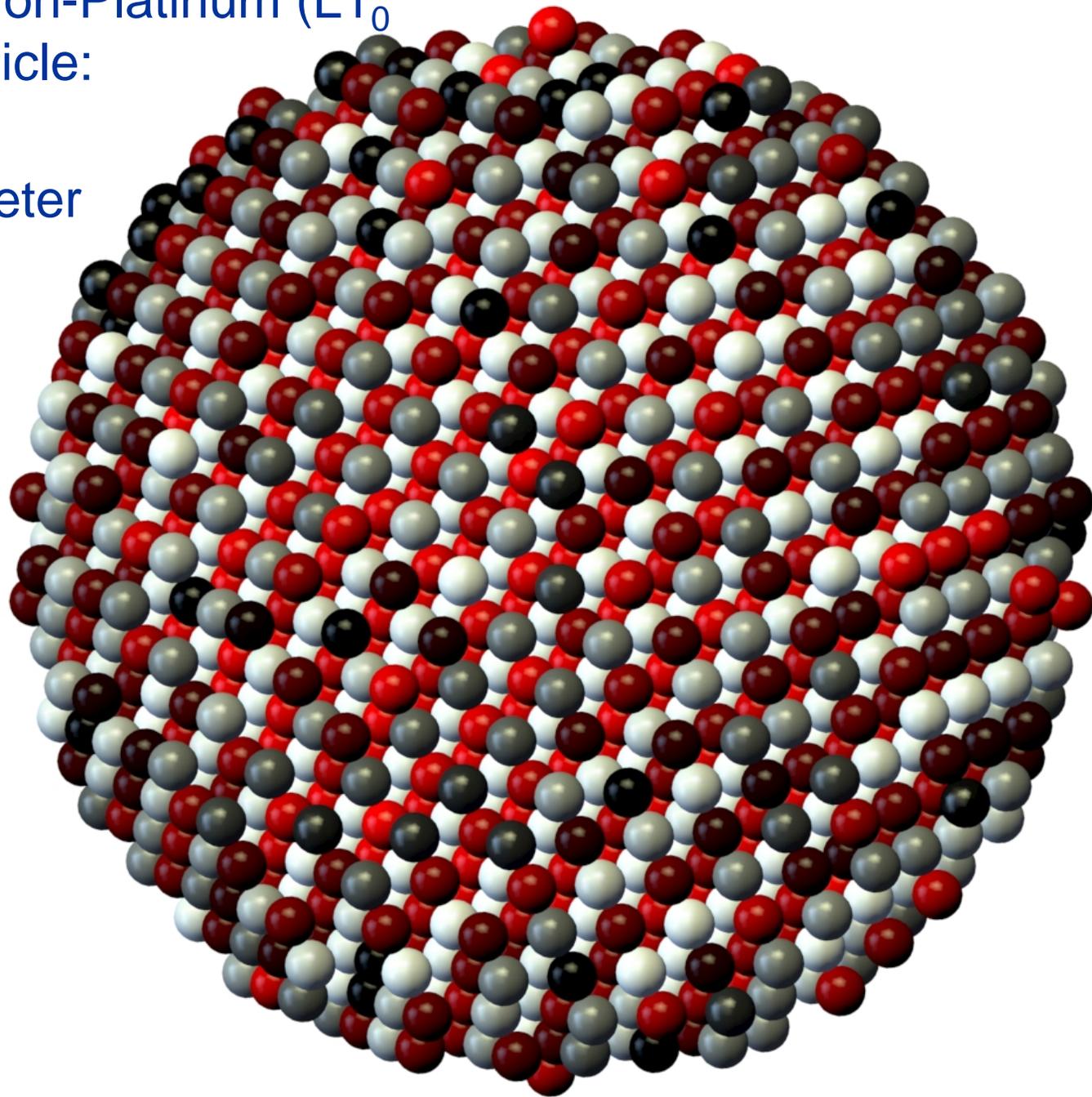
**Ordered compound
in $L1_0$ structure**

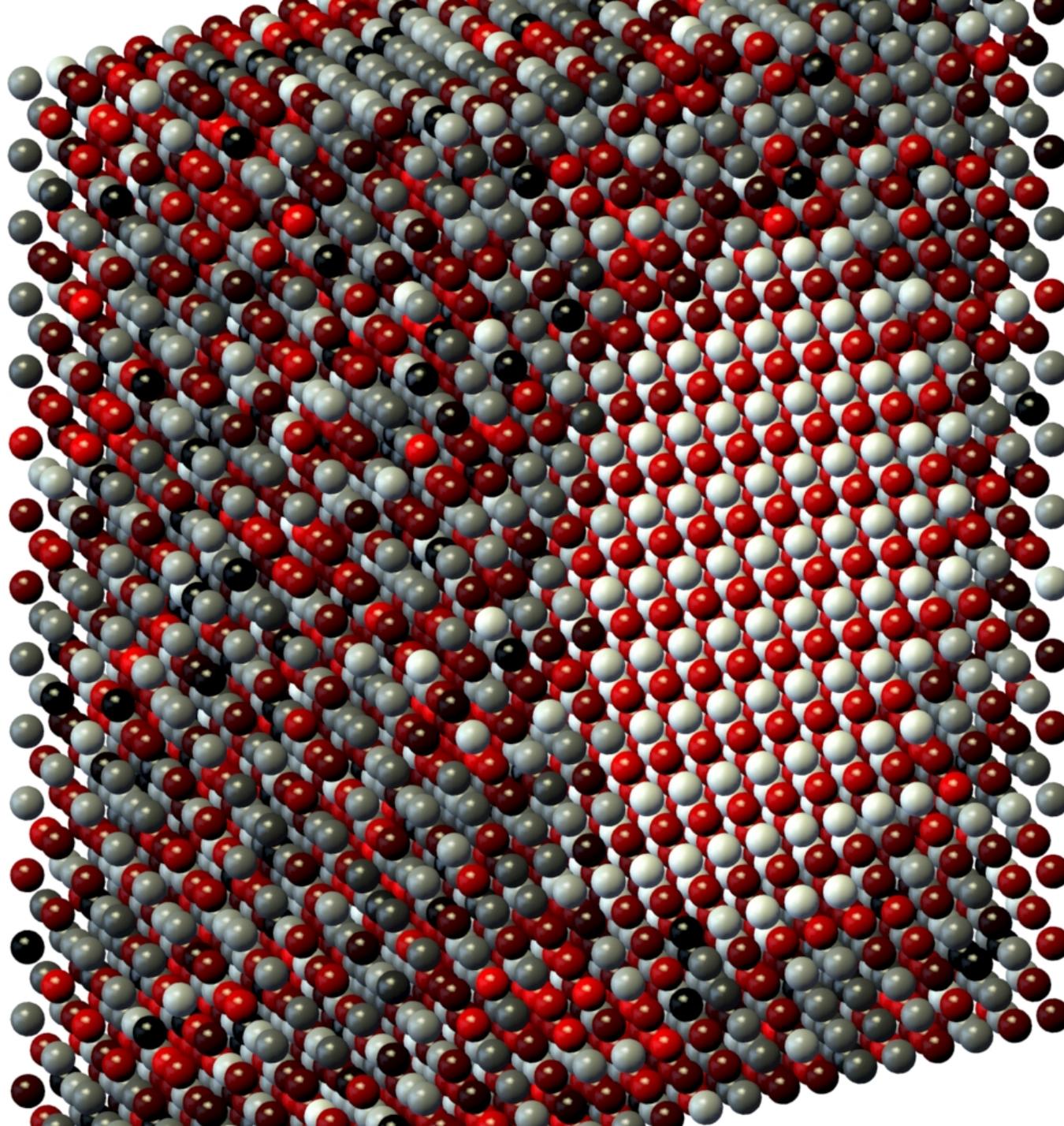


**Random alloy
in fct structure**



Ferromagnetic Iron-Platinum (L1₀
phase) nanoparticle:
2,195 atoms,
3.86 nm in diameter



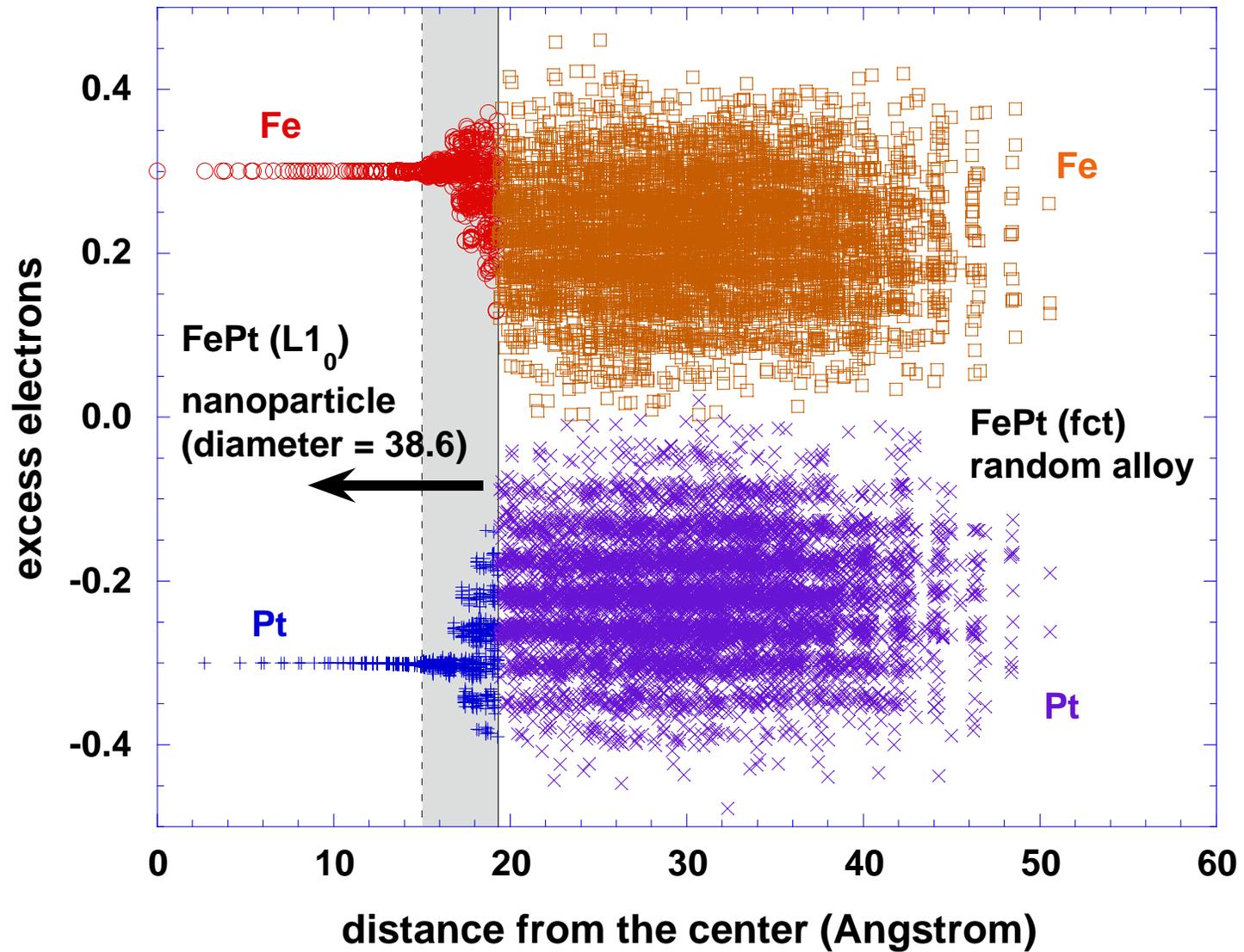


**Ferromagnetic
FePt nanoparticle
(3.86 nm)
embedded in FePt
fct random alloy**

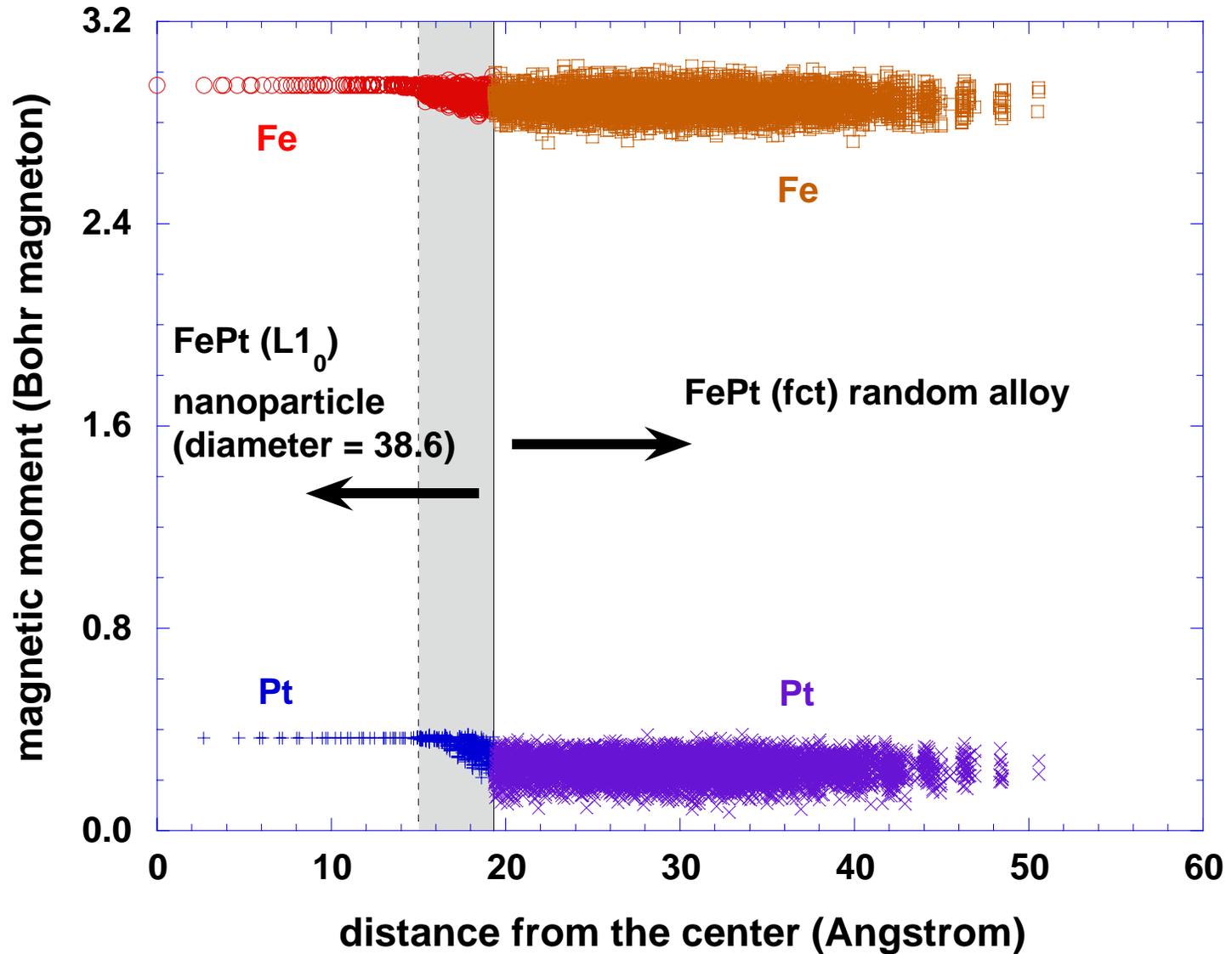
**Total simulation
size: 14,400 atoms**

By Greg Foss at PSC

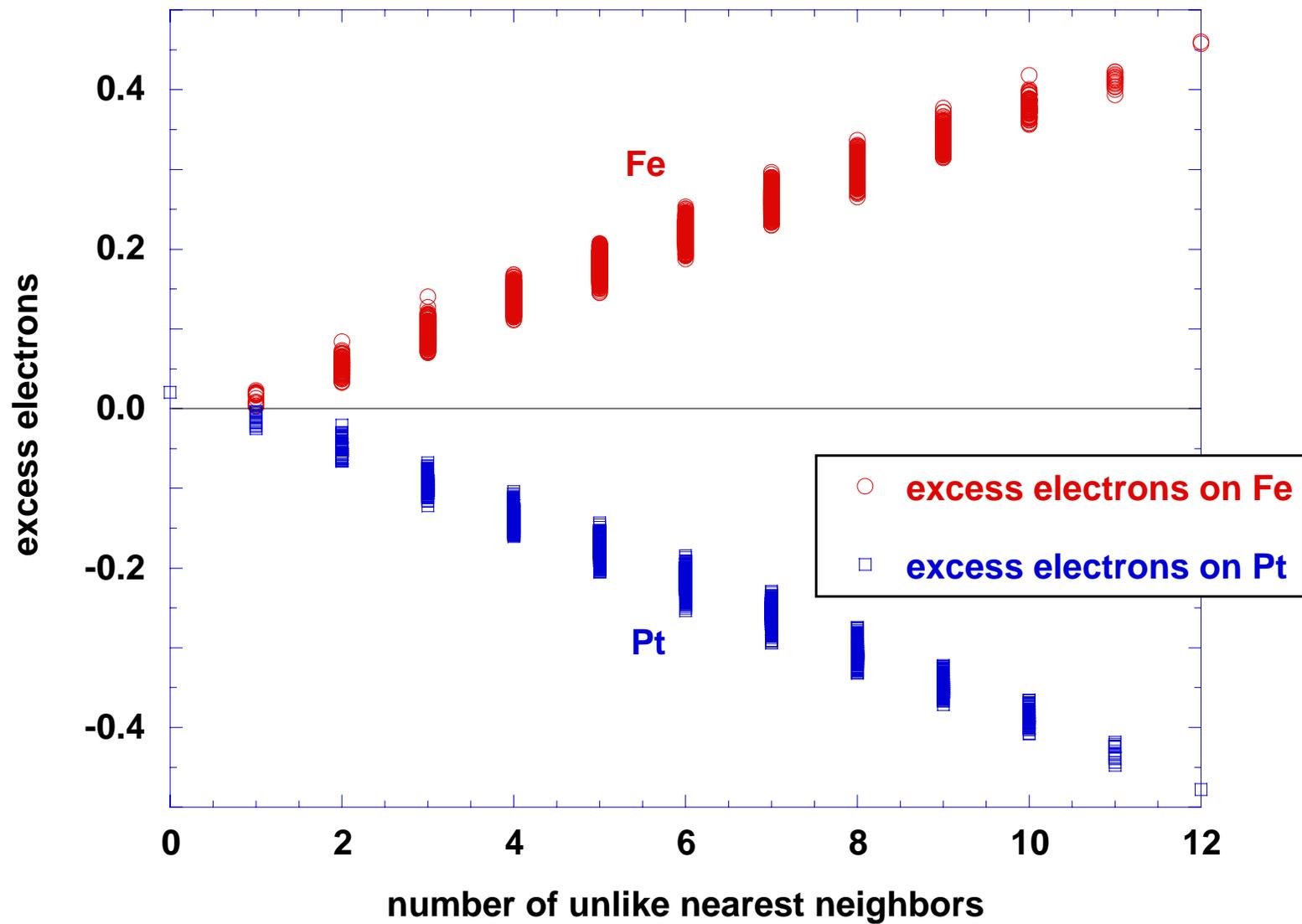
FePt nanoparticle embedded in FePt random alloy



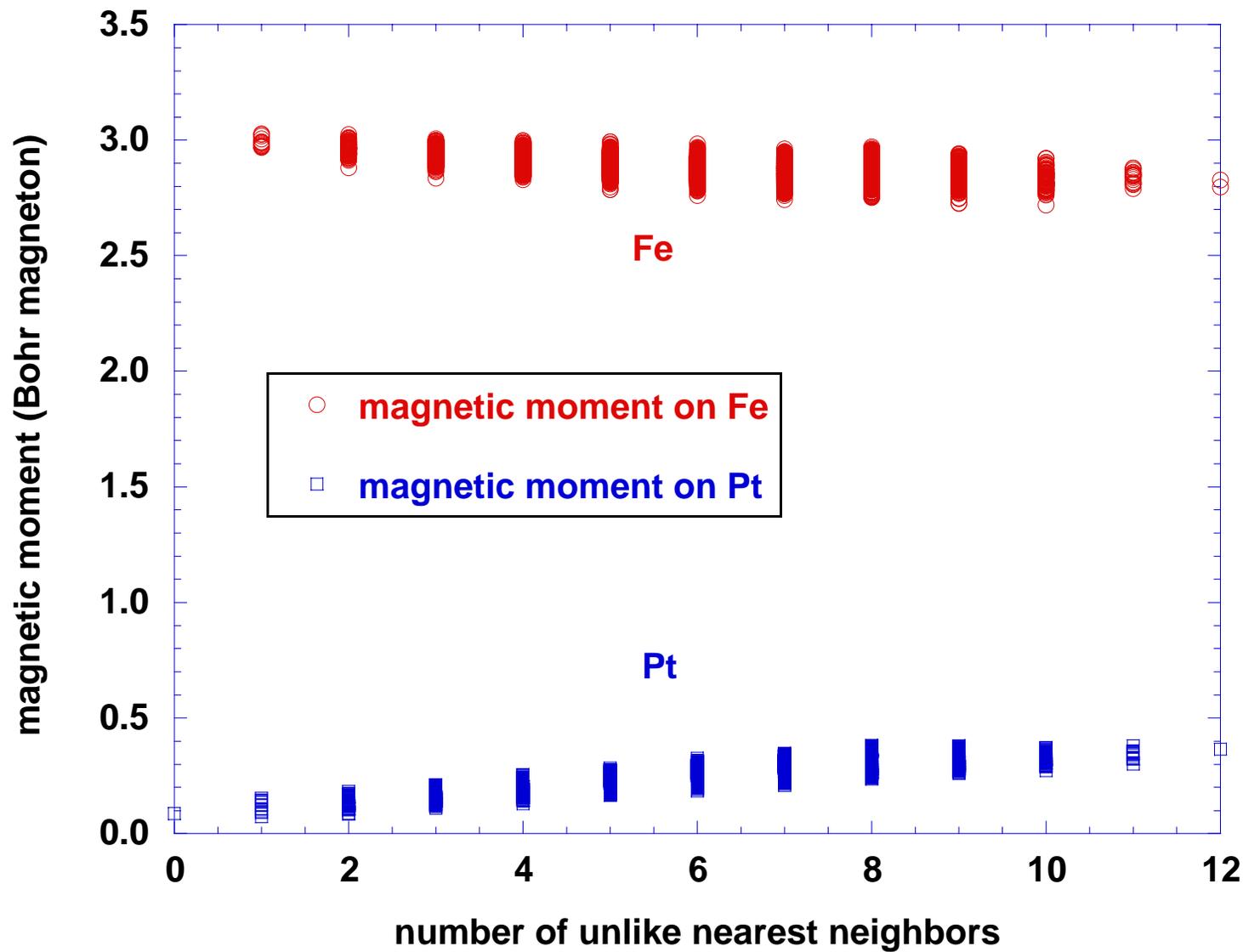
FePt nanoparticle embedded in FePt random alloy



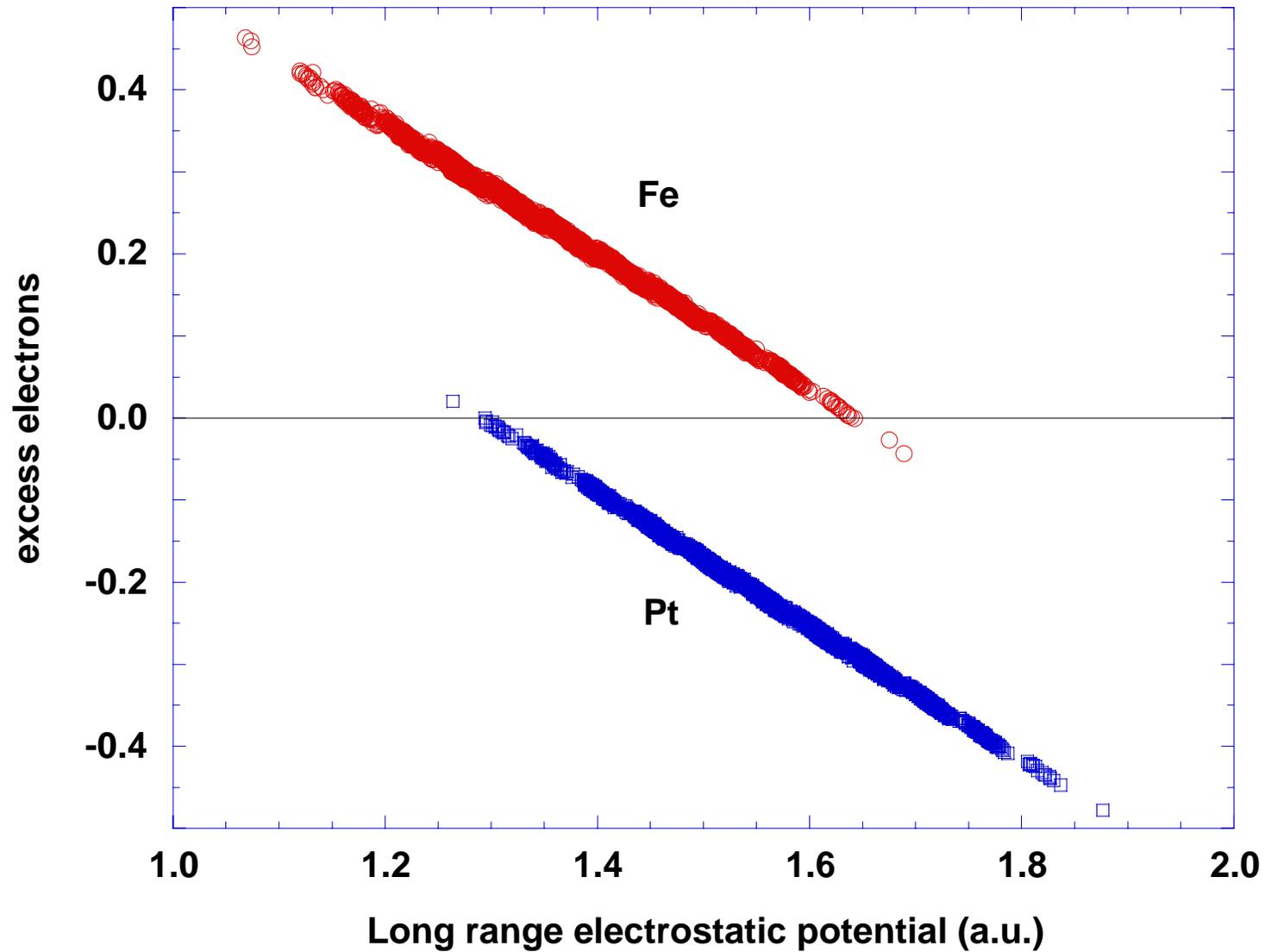
FePt nanoparticle embedded in FePt random alloy



FePt nanoparticle embedded in FePt random alloy



FePt nanoparticle embedded in FePt random alloy



Conclusions

- ❑ The central region of the nanoparticle resembles the bulk properties
- ❑ Dramatic changes from the bulk properties are seen in the surface region $\sim 4 \text{ \AA}$ in width
- ❑ The size effect on the charge and moment is small
- ❑ Strong correlations between the number of unlike nearest neighbors and the electronic and magnetic properties
- ❑ With the linear-scaling *ab initio* method and a 20 teraflop machine, we should be able to perform electronic structure calculations for nanoparticles of diameter size up to $\sim 10\text{nm}$ ($\sim 80,000$ atoms)
- ❑ It requires a petaflop machine to perform realistic simulations for nanoparticles of $\sim 50\text{nm}$ ($\sim 5,000,000$ atoms) in size, and other nanomaterials such as nanowire and nanotubes.

Questions to be Answered

- ❑ Interaction between the nanoparticles versus the distance, shape, etc
- ❑ Noncollinear magnetic structure at the interface region between the nanoparticles or between the nanoparticle and the surrounding matrix
- ❑ Size and shape effects on Magneto-anisotropy of the nanoparticle
- ❑ Relaxation in the interface region

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